

Growth Mechanism and Electronic Structure of Zn_3P_2 on the Ga-rich GaAs(001) Surface

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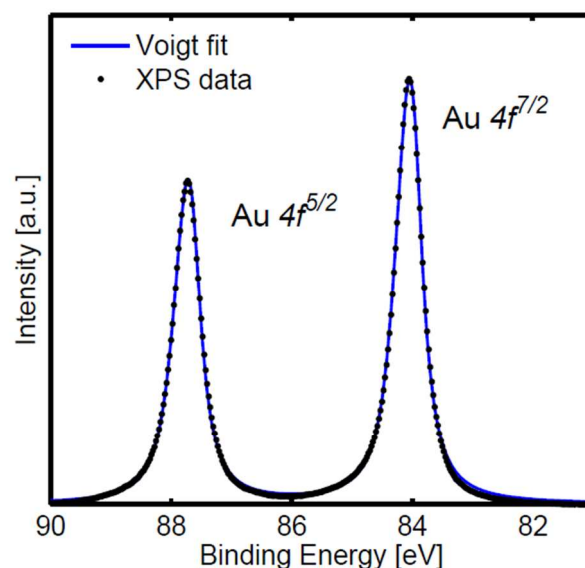


Figure S1. Au 4f core-level XPS spectra (black dots) and fitting (blue curve). Au 4f core-level XPS spectra were taken for reference from a 100 μm thick gold foil. The foil was cleaned with several cycles of sputtering (5 kV, Ar ion) and annealing at 650 $^{\circ}\text{C}$ in the Kratos system (background pressure 2E-9 torr). The pass energy and step size of the analyzer were 20 and 0.025 eV, respectively. The inherent lifetime broadening of the Au 4f core level is known to be 0.317 ± 0.010 eV [1], indicating that the Kratos instrument energy resolution was < 0.20 eV.

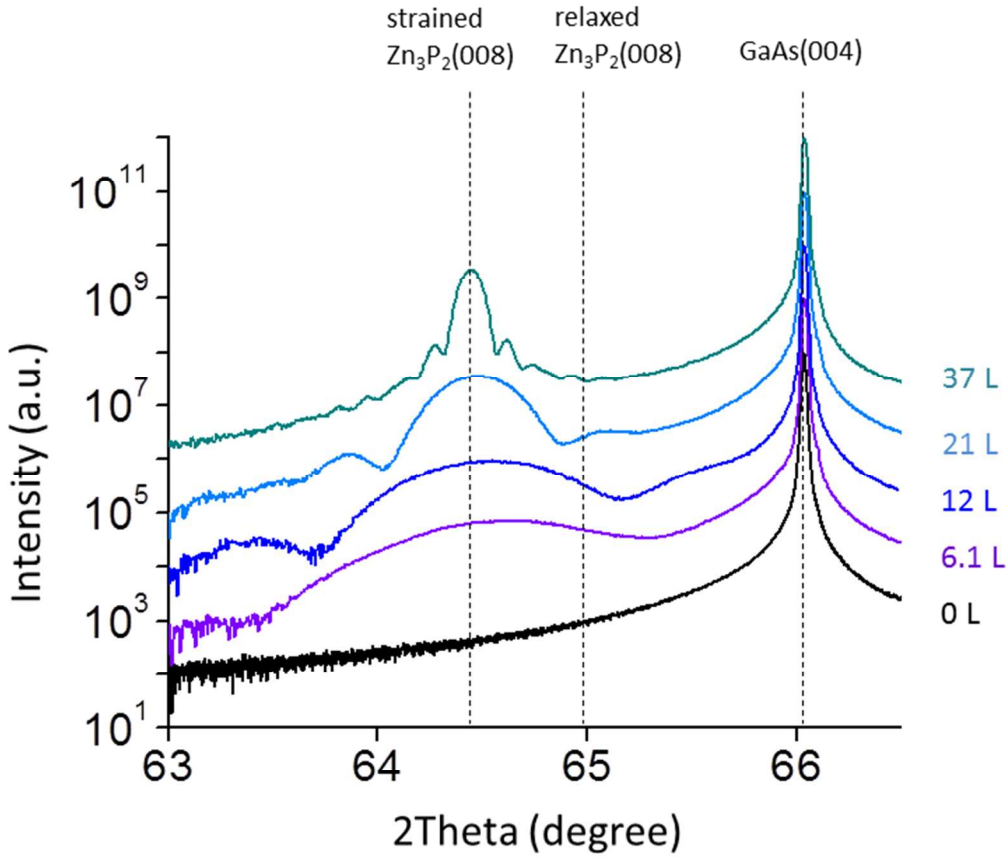


Figure S2. Symmetric HRXRD scans of the Zn₃P₂/GaAs(001) with the Zn₃P₂ deposition amount of 0, 6.1, 12, 21, and 37 L. Vertical dashed lines indicate the diffraction peak positions of pseudomorphic epitaxy Zn₃P₂ (008), crystalline Zn₃P₂ (008), and crystalline GaAs(004), respectively.

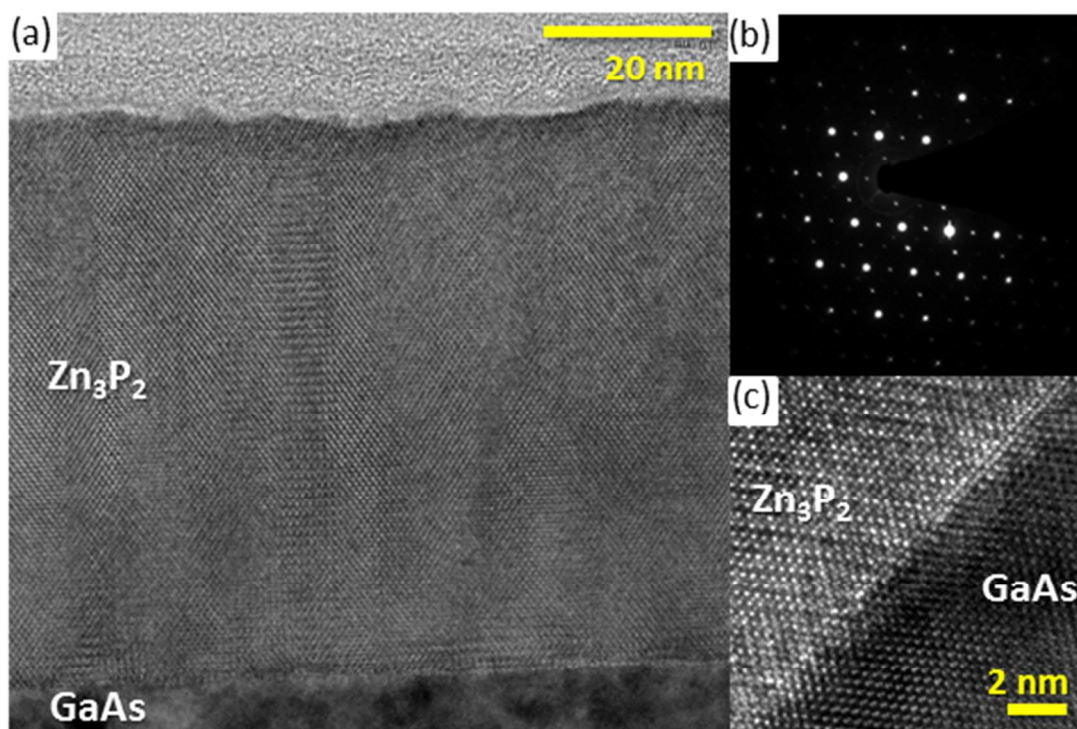


Figure S3. (a), (c) TEM images of the $\text{Zn}_3\text{P}_2/\text{GaAs}(001)$ with the Zn_3P_2 deposition amount of 37L. (b) SAED image of the $\text{Zn}_3\text{P}_2/\text{GaAs}(001)$.

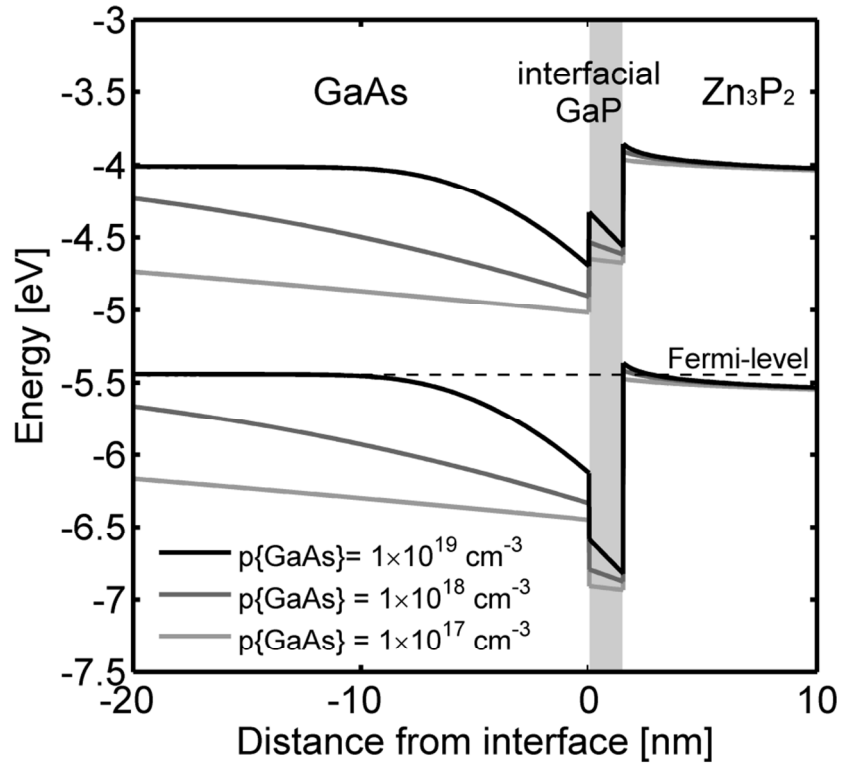


Figure S4. Calculate energy band diagrams of a $\text{Zn}_3\text{P}_2/\text{GaAs}(001)$ heterojunction with a 1.0 nm thick GaP interfacial layer as a function of the GaAs substrate hole concentration. The Zn_3P_2 hole concentration was fixed at $1 \times 10^{16} \text{ cm}^{-3}$ and the GaP interfacial layer was treated as intrinsic.

DFT calculation supplement

Surface formation energy of a ternary alloy is defined as,

$$E_{surf} = \frac{1}{A} [E(mnl) - m\mu(Ga) - n\mu(As) - l\mu(P)]$$

A : Area of surface

m , n , and l : Number of Ga, As, and P atoms, respectively

$E(mnl)$: Ab-initio total energy of a surface model including m of Ga atoms, n of As atoms, and l of P atoms

$\mu(B)$: Chemical potential of atom B (B = Ga, As, or P)

Change of surface formation energy with respect to a reference structure,

$$\Delta E_{surf} = \frac{1}{A} [E(mnl) - E_{ref}(m' n' l') + \mu(Ga)(m' - m) + \mu(As)(n' - n) + \mu(P)(l' - l)]$$

In our system,

- 1) The number of Ga atoms are maintained ($m' - m = 0$)
- 2) Anion replacement of As by P maintains total number of As and P ($n + l = n' + l'$)
- 3) $\Delta\mu(P)$ and $\Delta\mu(As)$ are defined as follows

$$\Delta\mu(P) = \mu(P) - \mu(P^{bulk}), (\Delta H_f(GaP) \leq \Delta\mu(P) \leq 0)$$

$$\Delta\mu(As) = \mu(As) - \mu(As^{bulk}), (\Delta H_f(GaAs) \leq \Delta\mu(As) \leq 0)$$

By substituting equations in 1) ~ 3) into ΔE_{surf} , we obtain,

$$\Delta E_{surf} = \frac{1}{A} \{E(mnl) - E_{ref}(m' n' l') +$$

$$(l' - l)[\mu(P^{bulk}) - \mu(As^{bulk}) + \Delta\mu(P) - \Delta\mu(As)]\}$$

$E_{ref}(m' n' l')$: Calculated total energy of the GaAs(001)- $\beta 2(2 \times 4)$ surface model including m' of Ga atoms, n' of As atoms, and l' of P atoms

$\mu(P^{bulk}) = -5.27575 \text{ eV}$: Calculated chemical potential of P from black phosphorus

$\mu(As^{bulk}) = -4.25974 \text{ eV}$: Calculated chemical potential of As from crystalline arsenic

$\Delta H_f(GaP) = -1.086 \text{ eV}$: Heat of formation of bulk GaP [2]

$\Delta H_f(GaAs) = -0.847 \text{ eV}$: Heat of formation of bulk GaAs [2]

REFERENCE

- (1) Kraut, E. A.; Grant, R. W.; Waldrop, J. R.; Kowalczyk, S. P. Semiconductor core-level to valence-band maximum binding-energy differences: Precise determination by x-ray photoelectron spectroscopy. *Phys. Rev. B* **1983**, 28, 1965.
- (2) *Landolt-Bornstein Numerical Data and Functional Relationships in Science and Technology - New Series, Group III: Crystal and Solid State Physics*; Springer-Verlag: Berlin, 1982, 17